

COMMUNICATION

First-Principles Study on the Cubic CaSiO_3 (001) Surface

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Abstract: The geometric and electronic structure of the cubic CaSiO_3 (001) surfaces have been studied using first-principles density functional theory (DFT) calculations. Two different terminations, CaO- and SiO_2 - terminated surfaces, were investigated. It has been found that Ca atom has the largest relaxation for both kinds of terminations, and the rumpling of the CaO-terminated surface is much larger than that of the SiO_2 -terminated surface. The band gaps of the CaO- and SiO_2 -terminated surfaces were calculated to be smaller than that of the CaSiO_3 bulk. It was also shown that the SiO_2 -terminated surface has a lower energy than the CaO-terminated surface.

AMS subject classifications: 68U05, 68U07

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1. Introduction

Composition estimates of the Earth reveal that the $\text{MgO-FeO-CaO-SiO}_2\text{-Al}_2\text{O}_3$ system could occupy ~ 99% of the mantle volume [1]. In the Earth's lower mantle, the Ca-bearing phase is present in the CaSiO_3 perovskite form [2,3], which is the third most important phase. Under ambient conditions, however, CaSiO_3 perovskite is not stable and it could readily convert to glass on the release of pressure. At 490-580 K and 27-72 GPa, CaSiO_3 perovskite undergoes

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phase transformation from tetragonal symmetry to ideal cubic structure [4].

Density functional theory (DFT) studies have been carried out on the lattice structure, sound velocity and elastic properties of bulk CaSiO_3 in tetragonal and orthorhombic phases as well as in low-symmetry cubic phase [5-7]. However, the structural and electronic properties of the CaSiO_3 surface are rarely investigated, despite the fact that the surfaces of other Ca-bearing perovskites, e.g. CaTiO_3 [8-10] and CaZrO_3 [11], have been well studied. In this work, first-principles calculations were carried out on the systematic study of the geometric and electronic properties of cubic CaSiO_3 (001) surface with CaO and SiO_2 terminations. The rumpling of the surfaces, band structures and energetic properties of the two kinds of terminations are compared to each other.

2. Computational details

The DFT calculations presented in this work were carried out within the generalized gradient approximation (GGA), using the projector-augmented wave (PAW) method and a plane-wave basis set, as implemented in the Vienna *ab initio* Simulation Package (VASP) [12,13]. The plane-wave energy cutoff is 600 eV for all calculations and the Brillouin zone integration is performed using the Monkhorst-Pack scheme with Ca (3s, 3p, 4s), Si (3s, 3p), and O(2s, 2p) treated as valence.

3. Results and discussion

3.1 Structural properties

We first optimized lattice constant of cubic bulk CaSiO_3 with the $12 \times 12 \times 12$ k -point mesh. The computed lattice constant of cubic CaSiO_3 is 3.604 Å, which is in good agreement with the equation of state (EOS) determined lattice constant of cubic CaSiO_3 in previous experimental study (3.572 Å) [14] and other GGA calculated result (3.546 Å) [5]. This theoretical lattice constant was used in all surface calculations presented here. Two symmetrical repeat-slab surface models with space group $P4/mmm$ were used for the calculations: CaO-terminated and SiO_2 -terminated surfaces. For the CaO-terminated surface, the unit slab consists of four CaO and three SiO_2 layers, so that the slab is terminated with CaO layer on either surface. Similarly, for the SiO_2 -terminated slab, there are three CaO and four SiO_2 layers in the unit slab, so SiO_2 layers are terminated on both outmost surfaces. For each unit slab, there are seven alternating CaO and SiO_2 layers, together with a 12 Å vacuum separation to minimize possible interactions between neighboring slab surfaces. For both slabs, the in-plane lattice constant is set to the computed cubic equilibrium value 3.604 Å, and the atomic

displacements perpendicular to the surface are fully relaxed using a $12 \times 12 \times 2$ k -point mesh.

Because of the P4/mmm symmetry, there is a mirror reflection on the central layer of each slab, thus the relaxations of the atoms on top and bottom layers are of the same amplitude but along opposite directions, and so are other symmetrical layers. **Table 1** lists the atomic displacements δz of the outmost three layers. For CaO-terminated surface, the surface layer Ca atom and O atom move toward opposite directions, i.e. the Ca atom moves inward to the central layer and the O atom relaxes outward to the vacuum region. The surface Ca atom has a large relaxation of 3.44% of the bulk lattice constant, which is the greatest relaxation of all atoms of both layers. For the SiO₂-terminated surface, the surface Si atom relaxes remarkably toward the central layer by 2.64 %. However, the largest relaxation is not on the surface layer atoms but on the second layer Ca atom, which relaxes outward (toward the surface) by 2.73%. It is noted that the Ca and Si atoms relax toward opposite directions with respect to the same-layer O atoms except the second layer of CaO-terminated slab. The Ca and Si atoms relax much more remarkably than O atoms in all layers. It is also found that the displacements of Ca atoms are much larger than those of Si atoms in both surface slabs, suggesting that Ca atom is much easier to relax than Si atom.

Table 1. Calculated atomic displacements (relative to ideal positions) δz of the top three layers of CaO- and SiO₂-terminated CaSiO₃ surfaces. Units are in percent of theoretical CaSiO₃ cubic bulk lattice constant ($a=3.604$ Å). Positive values refer to displacements towards the surface.

layer	CaO-terminated		SiO ₂ -terminated	
	Atom	δz	Atom	δz
1	Ca	-3.44	Si	-2.64
	O	0.79	O	1.35
2	Si	1.16	Ca	2.73
	O	0.03	O	-0.09
3	Ca	-0.45	Si	-0.35
	O	0.14	O	0.20

In **Table 2**, we show the surface relaxation parameters. Surface rumpling parameter s measures the outward displacement of the surface layer O atom with respect to the surface layer Ca or Si atom. Δd_{12} is the change of the first interlayer spacing, as measured from the surface to the second layer Ca and Si atoms z -coordinate, and Δd_{23} is similar with Δd_{12} but between the second and the third layers. It can be seen from **Table 2** that the rumplings of CaO-terminated and SiO₂-terminated surfaces are very close, i.e. the rumpling of CaO-terminated surface is only slightly larger than that of SiO₂-terminated surface. Thus, our calculation suggests that the CaSiO₃ (001) surface could be almost even rough when it is terminated with either CaO or SiO₂ termination. Parameter Δd_{12} is negative for both kinds of

terminations, indicating that the distance between the first and the second layers becomes smaller by comparison with its bulk value. The absolute value of Δd_{12} for SiO₂-terminated surface is larger than that of CaO-terminated surface, suggesting that the distance between the surface layer and the second layer is much more reduced for SiO₂-terminated surface. On the contrary, Δd_{23} , the distance between the second and the third layers expands for both terminations, and the expansion is larger for SiO₂-terminated surface than for CaO-terminated surface.

Table 2. Surface relaxation parameters for CaO- and SiO₂-terminated CaSiO₃ surface. Values are in percent of theoretical CaSiO₃ cubic bulk lattice constant ($a=3.604 \text{ \AA}$)

Surface	s	Δd_{12}	Δd_{23}
CaO-terminated	4.23	-4.60	1.61
SiO ₂ -terminated	3.99	-5.37	3.08

3.2 Band structures and partial density of states

In **Figure 1**, we show the calculated band structures of CaO- and SiO₂-terminated relaxed surfaces together with the band structure of bulk cubic CaSiO₃. The cubic bulk CaSiO₃ is calculated to have an indirect band gap of 3.55 eV, with the valance band maximum locating at R point and the conduction band minimum at Γ point as presented in **Figure 1(a)**. As can be seen from **Figure 1(b)**, the top valence band of CaO-terminated surface is flat between X and M points, with the calculated band gap of 2.57 eV. This gap value shows ~1 eV reduction with respect to its bulk band gap value. For the band structure of the SiO₂-terminated surface in **Figure 1(c)**, the calculated band gap is 1.62 eV, which is more reduced compared to its bulk value. The driving force for this relatively larger gap reduction is partially due to the tendency of the upper valence band states intruding upward near the M point. In addition, the lower conduction bands of SiO₂-terminated surface locate even lower than those of the CaO-terminated one, which narrows the band gap much more.

To clarify the reason of the reduced gaps of the surface slabs compared with the bulk, we then calculated the partial density of states (PDOS) of CaO- and SiO₂-terminated CaSiO₃ surfaces, as is shown in **Figure 2**. For both two surface types, O 2p states occupy the topmost valence bands, and Si 3s and 3p states contribute most to the lowest conduction bands. It is obvious that the central layer Si 3s and 3p PDOS of the CaO-termination is quite similar to the third layer Si 3s and 3p PDOS of the SiO₂-termination. However, the second layer Si 3s and 3p PDOS of the CaO-termination is very different to the surface layer Si 3s and 3p PDOS of the SiO₂-termination. The SiO₂-termination surface layer Si 3s and 3p has relatively large PDOS in the lowest conduction band area, ranging from 1.62 eV to ~4 eV, which suggests

that the surface Si 3s and 3p PDOS intrusion into the lower part is the main reason of the large band gap reduction of the SiO₂-terminated surface with respect to that of the CaSiO₃ bulk.

3.3 Surface energies

In order to compare the energetic stability of CaO-terminated and SiO₂-terminated surfaces, we calculated their surface energies using the standard approach of the surface energy calculation [8,11,15]. The surface energy E_s is defined as the sum of the cleavage energy (E_{cle}) and relaxation energy (E_{rel}):

$$E_s(I) = E_{cle} + E_{rel}(I) \quad (1)$$

where “I” denotes CaO or SiO₂ terminations. Since the two terminations form simultaneously under cleavage, it is assumed that the relevant cleavage energy is the same for both terminations:

$$E_{cle} = \frac{1}{4} [E_{slab}^{unrel}(CaO) + E_{slab}^{unrel}(SiO_2) - 7E_{bulk}] \quad (2)$$

Where $E_{slab}^{unrel}(CaO)$ and $E_{slab}^{unrel}(SiO_2)$ are the energies for unrelaxed CaO- and SiO₂-terminated slabs respectively, and E_{bulk} is the cubic CaSiO₃ bulk unit cell energy. The factor 1/4 is due to the fact that four surfaces are created during the cleavage, and the factor 7 is introduced because the two seven-layer slabs represent seven bulk unit cells. The relaxation energy is defined as the energy change after relaxation:

$$E_{rel}(I) = \frac{1}{2} [E_{slab}^{rel}(I) - E_{slab}^{unrel}(I)] \quad (3)$$

Where $E_{slab}^{rel}(I)$ is the slab energy after relaxation. Since both top and bottom surfaces of the

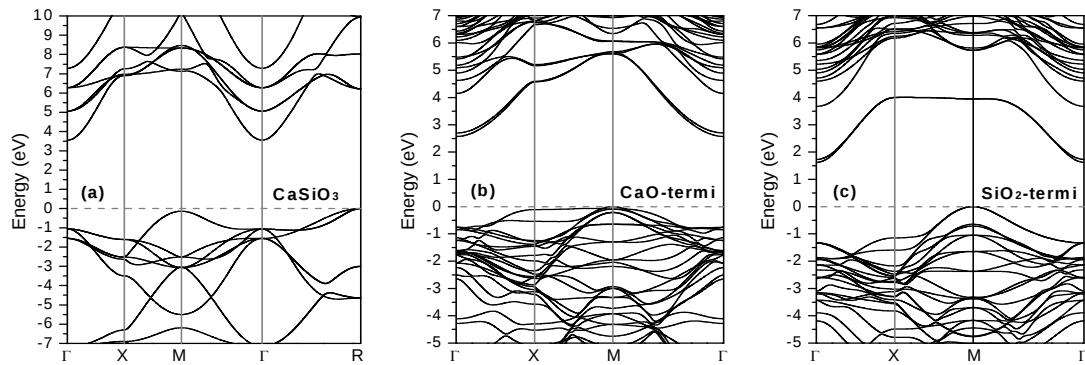


Figure 1: Band structures of CaSiO₃ cubic bulk (a), CaO-terminated surface (b) and SiO₂-terminated surface (c)

slab are relaxed, a factor 1/2 is introduced in Equation 3. The surface energies we obtained are $E_s(\text{CaO}) = 0.98\text{eV}$ and $E_s(\text{SiO}_2) = 0.87\text{eV}$, respectively, which indicates that the SiO_2 -terminated surface is energetically favorable. In other words, there could be more SiO_2 terminations on CaSiO_3 (001) surface rather than CaO terminated ones.

4. Conclusion

We have found from first-principles DFT calculations that for the cubic CaSiO_3 (001) surface, the rumpling of CaO -terminated surface is larger than that of the SiO_2 -terminated one. Ca atoms relax much more than Si and O atoms for both kinds of surface termination. Compared with the band gap of cubic bulk CaSiO_3 , the band gap of SiO_2 -terminated surface reduces much more than CaO termination, which is mainly due to the downward of the conduction band state of surface layer Si 3s and 3p states. Energy calculation reveals that the two models of surface terminations has relative close surface stabilities, with the SiO_2 -terminated surface more stable than the CaO -terminated one.

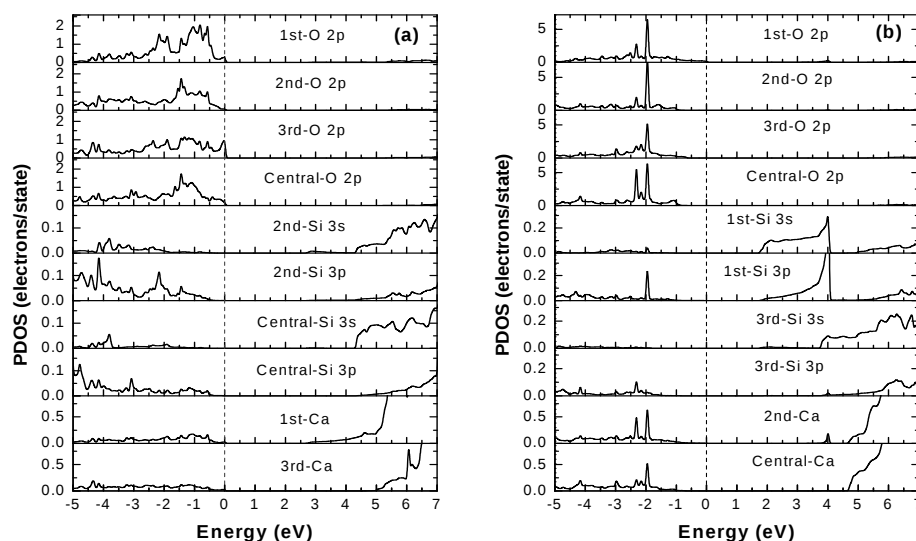


Figure 2: PDOS of atoms in each layer: (a) CaO -terminated surface and (b) SiO_2 -terminated surface.

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